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# Simulation of flow of short fiber suspensions through a planar contraction

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**Abstract** In this study, the flow of a fiber filled viscoelastic matrix through planar contractions is investigated. It was found that by adding fiber to the matrix vortex, the intensity increases. Fiber orientation along “x” and “y” axes was studied too. It was found that fiber orientation could be used for determining the flow regime through the contraction geometry. The rigidity condition of fibers, which needs the trace of the orientation tensor to be unity everywhere in the domain, is correct except near walls and the reentrant corner, which is slightly less than one. In these regions, the stress magnitude is higher, which results in more numerical errors, and which further leads to some error in predicting the orientation tensor. The first normal stress difference distribution along different axes was also studied in this article. It was found that increasing the volume concentration of fibers results in first normal stress difference intensification.

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## 1. Introduction

Contraction flow has attracted many studies, since it has very important industrial applications, such as extrusion and molding polymer melts [1]. Furthermore, it can be used for measuring the flow properties of Non-Newtonian fluids, such as polymer melts and polymeric solutions, which exhibit highly non-Newtonian behavior [2] and whose properties lie between purely viscous Newtonian fluids and completely elastic solid [3]. Consequently, more complicated constitutive equations, including Giesekus, Oldroyd-B, Phan-Thien-Tanner (PTT), FENE-P etc. have been used for evaluating the extra stress tensor. The complex structure of the polymer chain leads to some amazing phenomena, including die swell or Barus effect, tubeless siphon, rod climbing and elastic recoil [1–3]. The flow pattern in contraction geometry is very complex. Near the walls, shear flow dominates, while at the centerline, a purely elongational flow occurs [4].

Numerous studies have dealt with the contraction flow, both numerically and experimentally. In experimental investigations, some authors have used optical methods, such as Laser-Doppler Velocimetry (LDV), Flow Induced Birefringence (FIB) [5,6] and Digital Particle Image Velocimetry (DPIV) [7], in order to measure the velocity and stress fields.

There are many more numerical studies for both Newtonian [8] and non-Newtonian fluids [9–17]. For instance, Vrentas et al. [8] were the first to numerically study the creeping flow of a Newtonian fluid through an axisymmetric contraction.

In fiber suspensions, complex viscoelastic properties of a matrix, coupled with complexity, due to fiber-fiber, fiber-matrix and wall-fiber interactions, fiber re-orientation, fiber breakage and migration, result in very complicated rheological behavior for fiber-filled viscoelastic matrices. There are many numerical and experimental studies on the investigation and prediction of the rheological properties of fiber suspensions.

For a suspension of solid particles, the relative viscosity is mostly a function of the solid volume fraction, while for fiber suspensions, relative rheological functions are functions of the fiber volume fraction, the aspect ratio of fibers, fiber orientation and matrix properties [18]. It is worth mentioning that fiber diameter plays an important role in properties of fiber suspension, because during measurement of the rheological properties of suspension, fibers become aligned in the flow direction, such that the diameter would be the most important dimension of the fiber.

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Table 1: Viscoelastic matrix parameter used in simulation.

$\rho$ (kg/m <sup>3</sup> )	$\eta_s$ (kg/m.s)	$\eta_p$ (kg/m.s)	$\lambda$ (s)
1000	0.001	1.21	0.01

Flow characteristics of the suspension of the fiber in a thermoplastic media can be affected by the presence of fibers. On the other hand, when the molten polymer flows, fluctuating stresses modify fiber motion and orientation.

The first to study the flow of fiber suspension was Jeffery [19], followed by Batchelor [20], Cox and Brenner [21], and many more. However, most of these studies focus on dilute suspensions. Flow of a fiber suspension in contraction geometry has been studied by Lipscomb et al. [22] and Chiba and Nakamura [23]. Azaiez et al. [24] studied the flow of rigid fibers through a planar 4:1 contraction. They used three models, including PTT, FENE-CR and Carreau, for viscoelastic and Newtonian matrices. They found that fiber orientation is mostly affected by viscous effects. In places far enough from vortices, fiber orientation is not affected by matrix type, but near contraction walls, fibers are more orientated in polymer matrices. They also concluded that vortex intensity is reciprocally related to fiber orientation in the vortex region. An increase in fiber aspect ratio or concentration also results in a slightly more orientated state and a more intense vortex.

In this study, the flow dynamic and fiber orientation for flow of a fiber suspension in contraction geometry and fiber orientation are studied. A finite volume approach is used to solve constitutive equations of fiber suspensions in this geometry. The effect of numerical errors on predicting orientation tensor is investigated too.

## 2. Governing equations

In this study, the isothermal flow of a viscoelastic fluid containing fibers, through a planar contraction, is studied. Continuity and momentum equations for an incompressible viscoelastic matrix are as follows:

$$\nabla \cdot u = 0, \quad (1)$$

$$\frac{\partial(\rho u)}{\partial t} + \rho u \cdot \nabla u = -\nabla P + \nabla \cdot \tau^c, \quad (2)$$

where  $\rho$  denotes the fluid density,  $u$  is the velocity vector,  $p$  is pressure, and  $\tau^c$  is the composite stress tensor. Lipscomb et al. [22] have proposed a constitutive equation for dilute particle suspension. For ellipsoids with a high aspect ratio, the equation can be written as follows [25]:

$$\tau^c = \eta_m \gamma + \eta_m \psi \{ \mu_1 \gamma + \mu_2 \gamma : a_4 \}, \quad (3)$$

where  $\eta_m$  is the matrix viscosity that is calculated from matrix constitutive equations which is Newtonian or generalized Newtonian or viscoelastic. In this study, a viscoelastic matrix was considered with parameters shown in Table 1, where  $\eta_s$  and  $\eta_p$  are solvent and polymer viscosity and  $\lambda$  denotes the relaxation time. In Eq. (3),  $\psi$  denotes fiber volume concentration,  $\mu_1$  and  $\mu_2$  are rheological coefficients, and  $\gamma$  is the stress rate expressed by:

$$\gamma = \nabla u + \nabla u^t, \quad (4)$$

in which superscript “ $t$ ” denotes the transpose of the matrix. In Eq. (3), the last term is a representative of coupling between

fiber orientation and hydrodynamic forces. The expression  $\gamma : a_4$  is given by the quadratic closure approximation, given by:

$$\gamma : a_4 = \text{tr}(\gamma \cdot a) a, \quad (5)$$

in which “ $a_4$ ” and “ $a$ ” denote the fourth-order and second order orientation tensors, respectively, and “ $\text{tr}()$ ” denotes the trace of the tensor. There are several relation for  $\mu_1$  and  $\mu_2$ . For ellipsoids with large aspect ratio, this coefficient is given by Lipscomb et al. [22]:

$$\mu_1 = 2, \quad (6)$$

$$\mu_2 = \frac{r^2}{2 \ln(r)}, \quad (7)$$

where “ $r$ ” denotes the ellipsoids aspect ratio. Tensor “ $a$ ” is the orientation tensor for non-dilute suspension, given by Folgar and Tucker [26]:

$$\begin{aligned} \frac{da}{dt} = & -\frac{1}{2}(\omega \cdot a - a \cdot \omega) + \frac{1}{2}\xi(\dot{\gamma} \cdot a + a \cdot \dot{\gamma} - 2\dot{\gamma} : a_4) \\ & + 2C_I \bar{\gamma}(\delta - ma), \end{aligned} \quad (8)$$

in which, the last term describes the interaction between neighboring fibers. Parameter “ $m$ ” is 2 and 3 for two and three dimensional geometries, respectively. Tensor  $\delta$  is the identity tensor,  $\gamma$  is the shear rate tensor (Eq. (4)), and  $\omega$  is the vorticity tensor given by:

$$\omega = \nabla u - \nabla u^t. \quad (9)$$

In Eq. (8),  $\xi$  is given by:

$$\xi = \frac{r^2 - 1}{r^2 + 1}, \quad (10)$$

where “ $r$ ” is the ellipsoid aspect ratio:

$$r = \frac{L_f}{D_f}, \quad (11)$$

in which,  $L_f$  and  $D_f$  are length and diameter of the fibers, respectively. In Eq. (8),  $C_I$  is a phenomenological constant exhibiting fiber–fiber interactions. There are several relations for estimating  $C_I$ . In this paper, we used the following form:

$$C_I = 0.0184 \exp(-0.7148 \psi r). \quad (12)$$

In Eq. (12), “ $r$ ” is the fiber aspect ratio, given by Eq. (11).

Finally, these relations were implemented in the OpenFOAM software package. This software is open source software in which the user can implement new solvers and/or transport models in the source codes. The software uses a finite volume approach for solving the PDE set.

## 3. Geometry and mesh

A planar geometry, shown in Figure 1, is used for numerical simulations. The length was considered long enough to reach a fully developed condition. A non-uniform mesh is used for numerical simulation, which is exhibited in Figure 2. There are 47 700 cells in the used mesh in the simulation.

## 4. Results and discussion

A suspension of 10% volume of fibers with aspect ratio of 20 was investigated. Figure 3 shows the  $a_{11}$ , which shows the

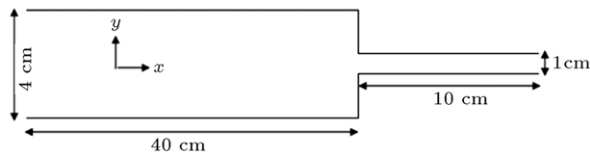


Figure 1: Geometry used for numerical simulation.

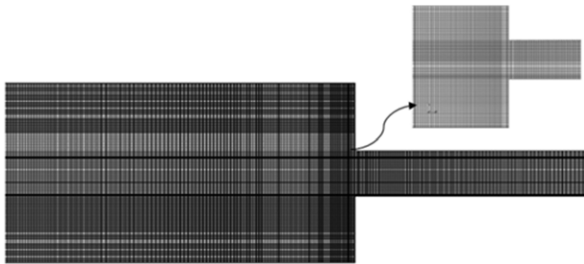
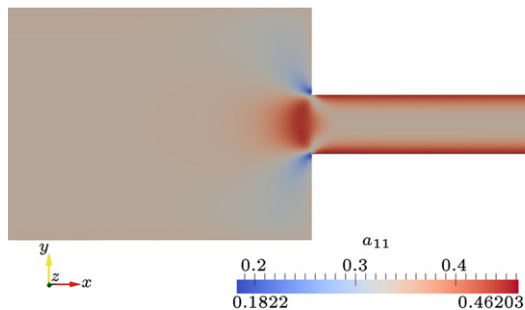
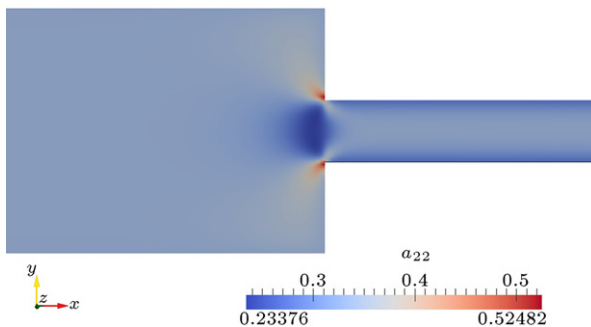


Figure 2: A non-uniform mesh used for numerical simulation.

Figure 3:  $a_{11}$  in part of contraction geometry.Figure 4:  $a_{22}$  in part of contraction geometry.

ellipsoid diameter along the  $x$  direction. It can clearly be seen that ellipsoids are more stretched near the die walls, because of shear stress exerting on fibers, and also just before the entrance region. In the entrance region, there is an elongational flow, which results in an enhancement in fiber orientation along the “ $x$ ” axis ( $a_{11}$ ). Next to the re-entrant corner, the  $a_{11}$  is small, because of fluid flow along the “ $y$ ” direction in this section (elongation along “ $y$ ”).

Figure 4 shows  $a_{22}$ , which is the representative of the ellipsoid diameter along the “ $y$ ” direction. As discussed before, for rigid fibers, the trace of tensor “ $a$ ” should be unity. So, increasing  $a_{11}$  results in a decrement in  $a_{22}$  and vice versa, as shown in Figures 3 and 4.

Figure 5 exhibits that limitation “ $trace(a) = 1$ ” is correct everywhere in the domain except at locations in which there

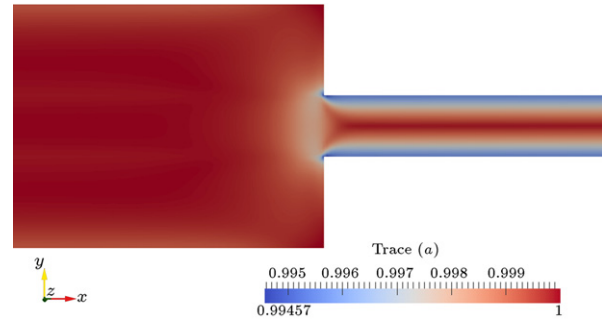
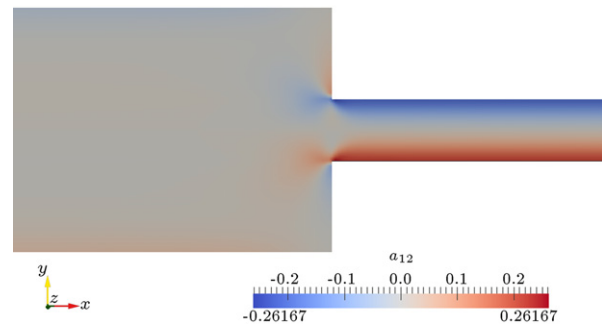


Figure 5: Trace of orientation tensor in the contraction geometry.

Figure 6:  $a_{12}$  in part of contraction geometry.

are intense stresses, such as at die walls in narrow sections. In these regions, numerical errors are bigger compared to inner sections, which leads to trace ( $a$ ) not being exactly unity, but near one.

Figure 6 shows the  $a_{12}$  component of the orientation tensor, which describes the angle between fiber axes and the “ $x$ ” direction. The symmetry in Figure 6 shows that fibers have an opposite orientation in the upper and lower sides of the symmetry plane. Positive values in the lower half denote that there is a positive angle between the fiber axis and the positive “ $x$ ” direction. Moreover, negative values for  $a_{12}$  in the upper half show that the fiber axis makes a negative angle with the “ $x$ ” direction.

For further discussion, it is needed to define two dimensionless lengths.  $X$  and  $Y$  are dimensionless lengths along  $x$  and  $y$  direction, respectively, and are given by:

$$X = \frac{(x - 0.4)}{0.01}, \quad (13)$$

$$Y = \frac{y}{0.01}. \quad (14)$$

The ellipsoid diameter along the “ $x$ ” direction ( $a_{11}$ ) at the centerline is plotted versus “ $X$ ”. Two different concentrations were considered. In places far from the entrance region, i.e.  $X < -4$  or  $X > 3$ ,  $a_{11} = 0.33$ , which means that the fiber has taken a spherical shape. Approaching the entrance region,  $a_{11}$  increases, which mean that fibers are aligned in an “ $x$ ” direction. In more concentrated solutions, there are more intense stresses, which lead the fiber to be more stretched. Thus, for more concentrated solutions,  $a_{11}$  is bigger in the same location.

Figure 8 shows the ellipsoids diameter along the “ $y$ ” direction. As mentioned earlier,  $a_{11}$  and  $a_{22}$  behave reversely, as can be observed from Figures 7 and 8.

Figure 9 exhibits the  $a_{12}$  profile along the centerline. Along the centerline, the value of  $a_{12}$  is very low, which means that

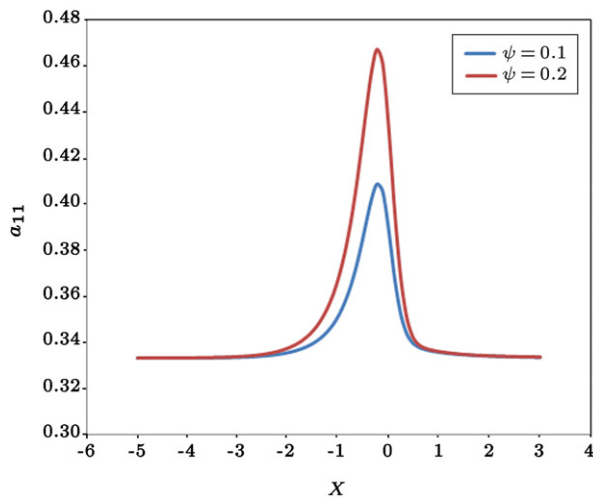


Figure 7: Ellipsoid diameter change along "x" direction at centerline for two suspensions with different volume concentrations.

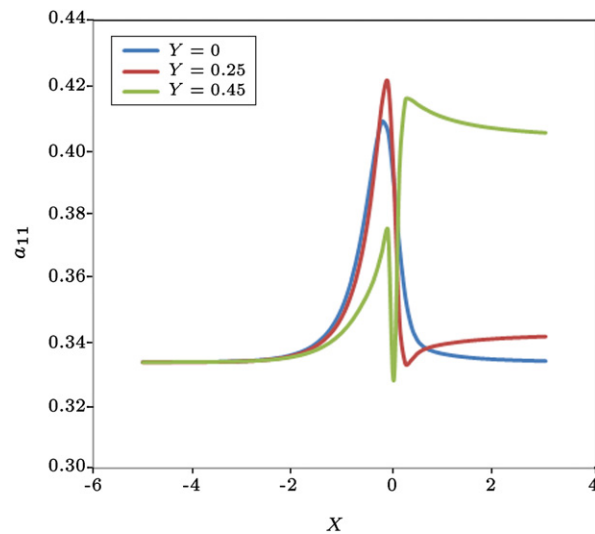


Figure 10:  $a_{11}$  for a suspension containing 10% fiber along "x" direction at different distances from the centerline.

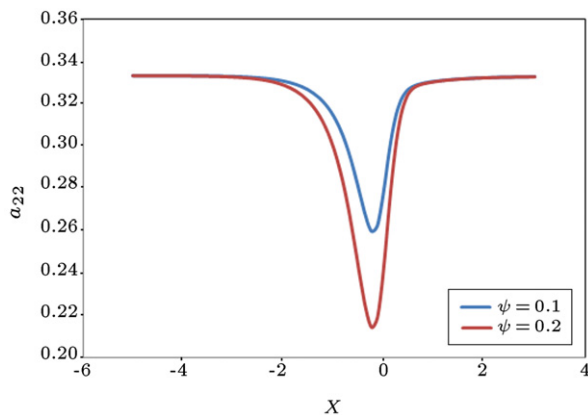


Figure 8: Ellipsoid diameter change along "y" direction,  $a_{22}$ , at centerline for two suspensions with different volume concentrations.

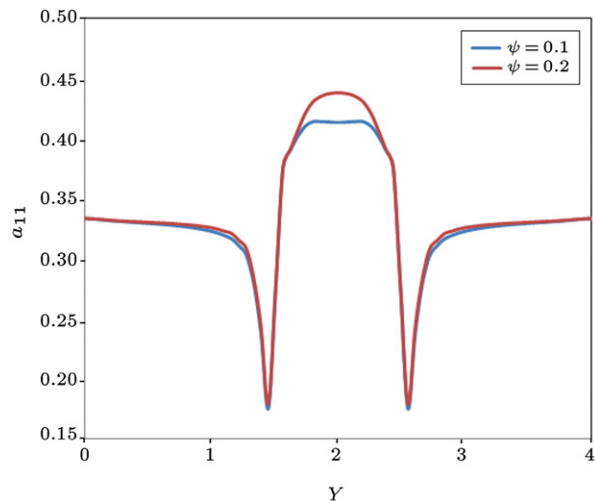


Figure 11:  $a_{11}$  at  $X = -0.03$  along "y" direction.

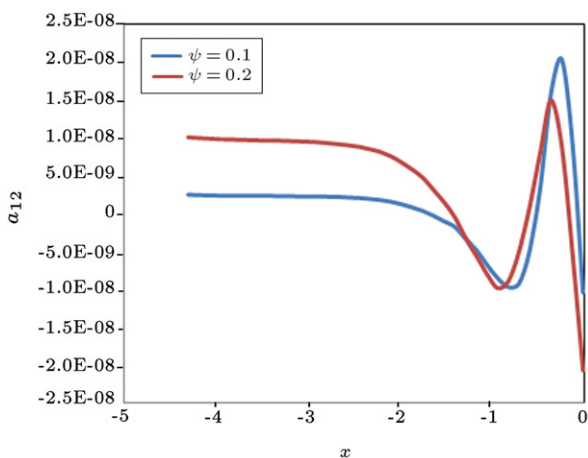


Figure 9:  $a_{12}$  change along "y" direction at centerline for two suspensions with different volume concentrations.

fibers are oriented along the "x" direction. There is a slight direction change just before the entrance ( $X > -2$ ).

Figure 10 exhibits  $a_{11}$  for a suspension containing 10% fiber, along the "x" direction, at different distances from the

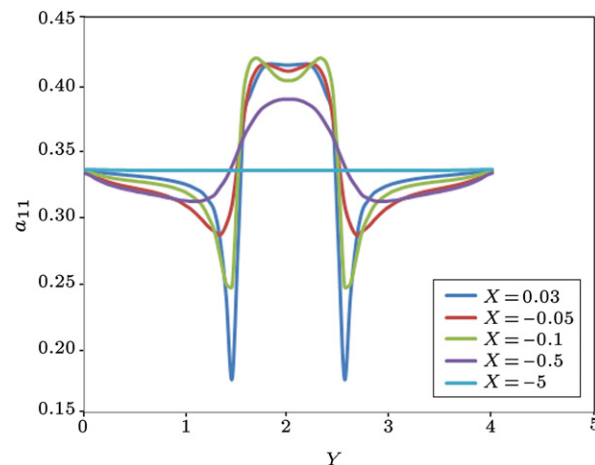


Figure 12:  $a_{11}$  at different distances from entrance along "y" direction.

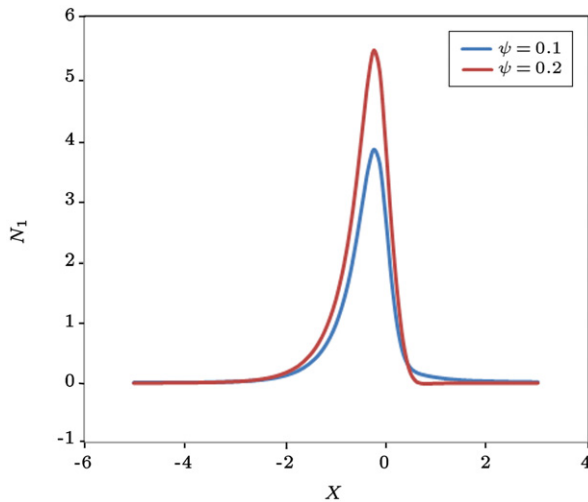
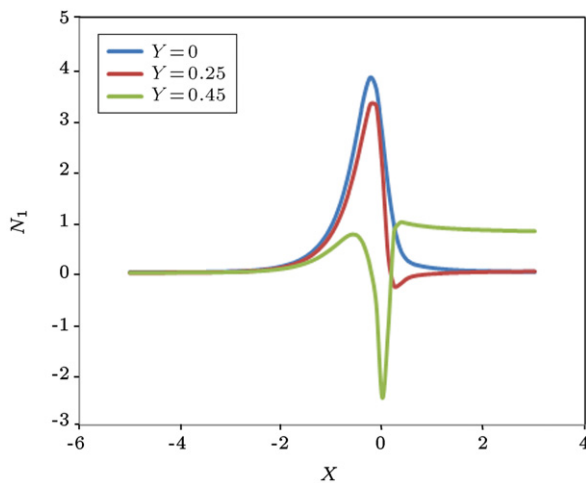
Figure 13: Normal stress distribution along centerline for  $Y = 0$ .

Figure 14: Normal stress distribution for different distances from centerline along "x" direction.

centerline. When "Y" tends to 0.5, such as  $Y = 0.45$ , there are two maximums and a minimum in the profile, while for smaller "Y", there is only a maximum. For  $Y = 0$  and  $Y = 0.25$ , the maximum value for  $a_{11}$  shows the region near the entrance where elongational flow dominates. For  $Y = 0.45$ , the first maximum occurs approximately at the same X, followed by a minimum, which indicates the region of narrow die, where a side flow enters from the re-entrant corner, which is perpendicular to the main flow. Then the second maximum occurs in the narrow die, where there are more intense stresses near the die wall.

In other words, two maximum are related to just before entering the narrow die and at the die wall, and the minimum corresponds to a small region in the die that begins just after the re-entrant corner.

Figure 11 shows  $a_{11}$  along y direction for different concentrations in  $X = -0.03$ . The figure shows that there is a maximum at the center and two symmetric minimum near the re-entrant corner. There is a region in the center where aurcelate, like curvature, is observed, in which there is an intense elongational flow. But, next to this region, shear flow dominates, which leads to a linear increase in  $a_{11}$ .

Figure 12 shows  $a_{11}$  along y direction at different distances from the die entrance for a suspension of 10% volume of fiber in the matrix. At very distant locations, ellipsoids are at rest, and diagonal components are all  $1/3$ , as shown for  $X = 5$ . Approaching the entrance, it is interesting that for places near the entrance region, some local recycling slight flow seems to occur, which leads to a minimum in the bell shaped section of the curve, as shown for  $X = 0.1$  in Figure 12.

Figure 13 exhibits that the first normal stress difference is higher for suspension with a higher volume concentration of fibers.

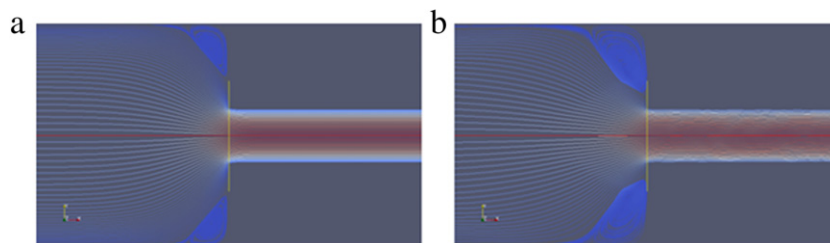
Figure 14 exhibits a first normal stress difference along the x direction at different Y.

Figure 15 compares the vortex size for two suspensions of different fiber concentrations. As illustrated, vortex size increases with fiber concentration in the solution.

## 5. Conclusions

In this study, the flow of a viscoelastic matrix filled with different volume concentrations of rigid fibers in 4:1 planar contractions was investigated, using the OpenFOAM software package. The trace of tensor "a" should remain unity for rigid fibers, which was confirmed by calculating the trace of tensor "a" everywhere in the domain. It was found that vortex size increases with enhancing the volume concentration of fibers. For fiber motion along the "x" direction near the centerline, it was found that ellipsoid diameter along the "x" axis or  $a_{11}$  reaches a maximum just before entering the abrupt contraction, which corresponds to a minimum in  $a_{22}$  at the same point. However, if one considers fiber orientation along the x direction near the narrow section walls, two maximums and one minimum are observed, where the minimum occurs because of flow entering from the re-entrant corner in the "y" direction, which is perpendicular to the main flow direction, and causes a local traverse direction.

In the "y" direction near the entrance region, one maximum and two minimums are observed for  $a_{11}$  and the maximum is bigger for more concentrated suspensions. It was found that the fiber orientation can be used in determining flow type, i.e. where the shear flow or extensional flow dominates.

Figure 15: Vortex growth with increasing fiber volume concentrations. (a)  $\psi = 0.01$ ; and (b)  $\psi = 0.16$ .



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